

## Nitrate Reduction

# Low-Valent Single-Atom Indium Site Regulating Ionic Interference and Adsorbed Hydrogen for Near-Unity Electrosynthesis of Ammonia

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**Abstract:** Microenvironment modulation, involving the selective adsorption of ions and the engineering of hydrogen radicals, is critical for the neutral electrochemical reduction of nitrate to ammonia at high current densities. In this work, self-adaptive low-valent indium single atoms SAs decorated copper-based nanosheets were investigated as a prototype. The catalyst exhibits a maximum ammonia Faradaic efficiency ( $FE_{NH_3}$ ) of 99.36% and a high  $NH_3$  yield rate of  $29.02 \text{ mg h}^{-1} \text{ mg}_{\text{cat}}^{-1}$  in neutral electrolyte. In-depth experiments and theoretical calculations suggest that the indium SAs optimize the local electronic distribution of the derived Cu matrix through strong p-d orbital couplings, with the electron-relay effect, thereby enhancing electron transfer and regulating the supply of hydrogen radicals to accelerate the hydrogenation process. Furthermore, in situ Raman results and molecular dynamics simulations reveal that the indium SAs can act as solid-state buffering sites by inducing a potential-dependent adsorption behavior of  $NO_3^-$  over  $SO_4^{2-}$  as a supporting oxoanion in the electric double layer, consequently maintaining high reaction activity and selectivity. Herein, the as-designed electrode operates stably at  $200 \text{ mA cm}^{-2}$  for 150 h in a bipolar membrane electrode assembly electrolyzer with a  $FE_{NH_3}$  of  $\sim 83\%$ , indicating promising practical applications.

## Introduction

Ammonia ( $NH_3$ ) is a vital feedstock in the industrial production of fertilizers, synthetic fibers, and pharmaceuticals.<sup>[1]</sup> To date, industrial-scale  $NH_3$  production still relies on the

Haber–Bosch technology, an energy-intensive process characterized by high temperatures and pressures, which is accompanied by significant carbon dioxide emissions.<sup>[2]</sup> Furthermore, the increasing levels of anthropogenic nitrogen fixation have disrupted the natural nitrogen cycle on Earth, leading to the accumulation of harmful nitrate ( $NO_3^-$ ) contamination in groundwater and posing significant environmental and health risks. Recently, eco-friendly electrochemical nitrate reduction reaction ( $NO_3^-$ -RR) has shown significant potential for achieving sustainable  $NH_3$  production and wastewater purification at ambient conditions.<sup>[3]</sup> Fundamentally, the electrosynthesis of  $NH_3$  is a complex proton-coupled electron transfer cascade process involving eight electrons and water dissociation to form adsorbed hydrogen radical (\*H) in neutral and alkaline medium, followed by the sequential hydrogenation of adsorbed  $NO_3^-$  and intermediates,<sup>[4]</sup> thus requiring continued research into the advanced catalyst design and meticulous microenvironment modulation.<sup>[5]</sup>

The  $NH_3$  yield rate and selectivity are primarily determined by the dynamic equilibrium between \*H generation and consumption, related to the coupling behavior with nitrogenous intermediates or other \*H in the  $NO_3^-$ -RR processes.<sup>[6]</sup> For instance, benefiting from the highly occupied d-orbital electrons, transition metal electrocatalysts have been widely explored, especially for Cu-based nanomaterials, which exhibit a superior affinity for  $NO_3^-$  ions.<sup>[7–12]</sup> However, this feature concomitantly enhances the \*H binding, aggravating the parasitic hydrogen evolution reaction (HER). In this regard, p-block metal elements with weak \*H binding are investigated as hetero-dopants or modifiers to manipulate the generation and coverage of \*H over Cu-based catalysts,

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steering the specific hydrogenation pathway.<sup>[13]</sup> Nevertheless, the effective dynamics control of the \*H is still limited due to the rapid consumption of  $\text{NO}_3^-$  ions at high reaction rate, resulting in severe concentration polarization of the electric double layer (EDL).<sup>[14]</sup> Especially in neutral wastewater, the  $\text{NO}_3^-$ -RR is constrained by the weak adsorption affinity of  $\text{NO}_3^-$  ions due to the electrostatic repulsion at the electrolyte-electrode interface and the coexisting competitive anionic adsorption with the supporting ions,<sup>[15]</sup> thus affecting the kinetics and pathways of  $\text{NO}_3^-$ -RR.<sup>[16]</sup> For instance, sulfate ( $\text{SO}_4^{2-}$ ) ions are common components in industrial effluents. They are widely applied as supporting electrolytes in neutral  $\text{NO}_3^-$ -RR, which can induce negative ionic interference in the space of the EDL. Currently, the synergistic regulation of \*H intermediate and the selective adsorption of  $\text{NO}_3^-$  in microenvironments remains underexplored.

In this work, as a proof of concept, we first synthesized p-block In single atoms (SAs) dispersed in ultrathin cuprous oxide nanosheets ( $\text{Cu}_2\text{O-In}_{\text{SA}}$ ) through in situ topological reconstruction of bimetallic coordination polymer nanosheets. In a neutral medium using  $\text{KNO}_3$  and  $\text{Na}_2\text{SO}_4$ , the obtained catalyst can achieve a maximum  $\text{NH}_3$  Faradaic efficiency ( $\text{FE}_{\text{NH}_3}$ ) of 99.36%, an  $\text{NH}_3$  yield rate of 29.02  $\text{mg h}^{-1} \text{mg}_{\text{cat}}^{-1}$ , and excellent electrocatalytic stability over 30 cycles for the  $\text{NO}_3^-$ -RR. In situ and ex situ experiments, as well as theoretical calculations, highlight that the low-valent In SAs, when decorating the oxide-derived Cu nanosheets with strong p-d orbital couplings through self-adaptive reconstruction of  $\text{Cu}_2\text{O-In}_{\text{SA}}$ , can enhance the electron transfer, favor the supply of \*H, and lower the energy barrier of the rate-determining protonation step. Meanwhile, the in situ Raman spectra and molecular dynamics (MD) simulations confirm that the In SAs in the oxide-derived Cu matrix can act as a solid-state buffer to effectively intensify the interfacial adsorption of  $\text{NO}_3^-$  ions by repelling the  $\text{SO}_4^{2-}$  ions as the  $\text{NO}_3^-$ -RR reaction proceeds with continuous  $\text{NO}_3^-$  ions consumption, further ensuring the high efficiency of ammonia production during the  $\text{NO}_3^-$ -RR. In this regard, the as-obtained electrocatalyst can operate stably at 200  $\text{mA cm}^{-2}$  for 150 h by applying a bipolar membrane electrode assembly electrolyzer with a sustained  $\text{FE}_{\text{NH}_3}$  of ~83%. It can also be efficiently driven by a solar cell under sunlight irradiation for  $\text{NH}_3$  synthesis, demonstrating its promising practical application potential.

## Results and Discussion

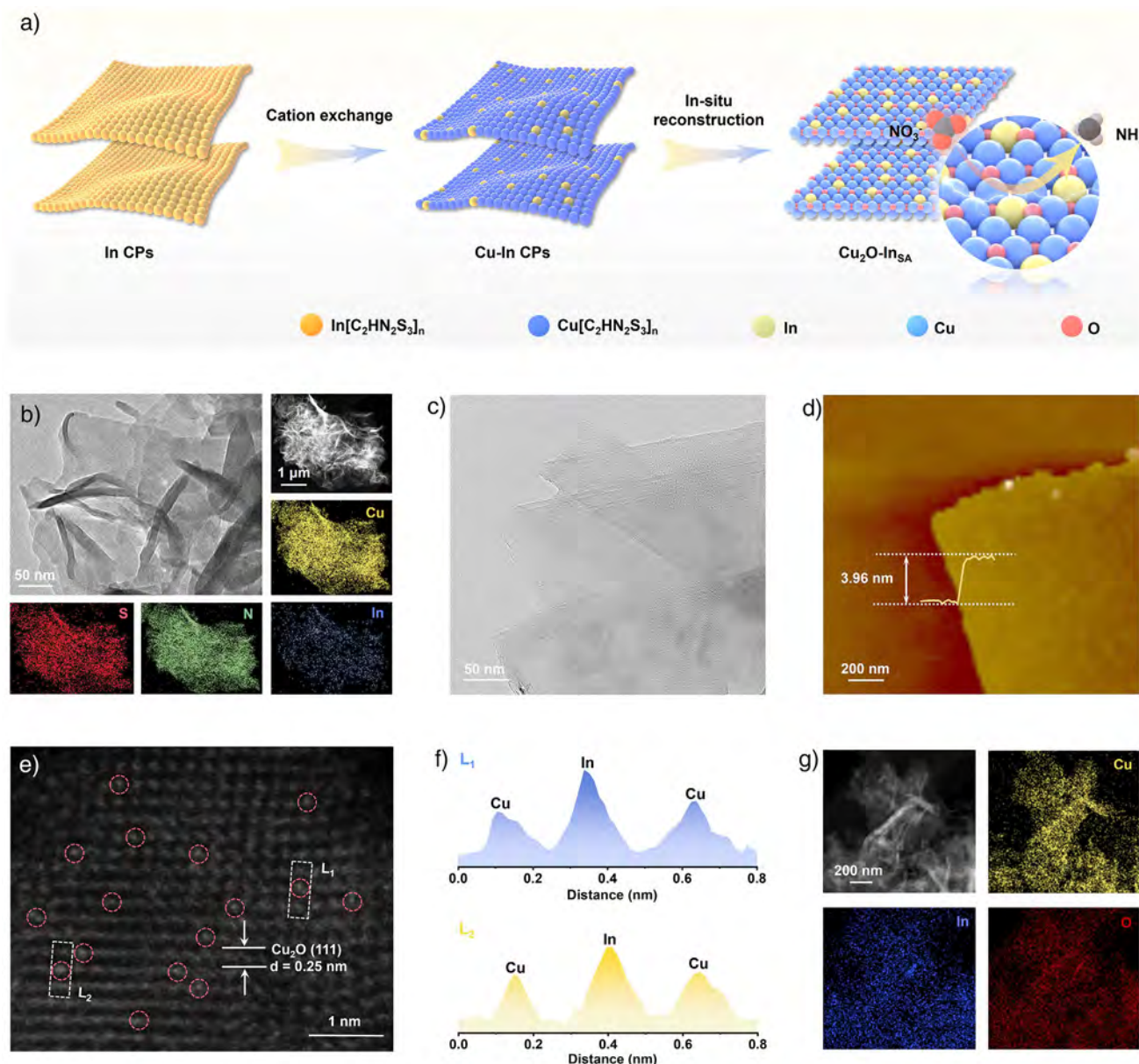
### Catalyst Synthesis and Characterization

The synthesis process of  $\text{Cu}_2\text{O-In}_{\text{SA}}$  is illustrated in Figure 1a, which involves a cation exchange reaction followed by an in situ electrochemical reconstruction process. First, the 2D indium coordination polymers (In CPs) nanosheets were synthesized with bismuthiol I ( $\text{C}_2\text{H}_2\text{N}_2\text{S}_3$ ) as ligand units through a weak reductant-induced polymerization process (Figure S1).<sup>[17]</sup> The In atom centers were then controllably substituted by  $\text{Cu}^{2+}$  cations to synthesize the Cu-In bimetallic coordination polymers (denoted as Cu-In CPs).

The corresponding transmission electron microscopy (TEM) image and energy-dispersive X-ray spectra (EDS) mapping reveal the well-dispersed Cu, In, N, and S elements in the Cu-In CPs nanosheets (Figure 1b). The atomic force microscopy (AFM) image discloses a thickness of ~4.3 nm for Cu-In CPs nanosheets (Figure S2). Afterward, the In SAs embedded  $\text{Cu}_2\text{O}$  nanosheets were synthesized via electrochemistry-induced topological reconstruction of the Cu-In CPs with ligand unit detachments. The  $\text{Cu}_2\text{O-In}_{\text{SA}}$  sample well maintains the 2D structure and exhibits apparent transparency under electron beam irradiation (Figure 1c). The AFM image confirms the ultrathin structure of  $\text{Cu}_2\text{O-In}_{\text{SA}}$  with a thickness of ~4.0 nm (Figure 1d). To probe the detailed structure at the atomic scale, an aberration-corrected high-angle annular dark field-scanning transmission electron microscopy (AC-HAADF-STEM) image was collected on the  $\text{Cu}_2\text{O-In}_{\text{SA}}$  nanosheets (Figure 1e). The discerned lattice spacings of 0.25 nm can be assigned to the  $\text{Cu}_2\text{O}$  (111) facets. Meanwhile, the brighter dots marked by red circles indicate the presence of monodispersed In SAs.<sup>[18]</sup> Moreover, the existence of isolated In atoms was further confirmed by the profiles along the dashed rectangles in the AC-HAADF-STEM image, where different atomic intensities are clearly distinguished (Figure 1f). EDS elemental mapping images also depict a homogeneous distribution of Cu, In, and O atoms throughout the entire  $\text{Cu}_2\text{O-In}_{\text{SA}}$  sample (Figure 1g).

The component of bimetallic coordination polymer precursors can be modulated by controlling the degree of cation exchange (Figure 2a). As revealed by Raman spectra, the typical peak around 167  $\text{cm}^{-1}$ , ascribed to the  $\text{A}_g^1$  vibration of the In-S bond in In CPs, is decreased in In-Cu CPs; then, this peak disappears, and the typical Cu-S signal at around 470  $\text{cm}^{-1}$  is generated in Cu-In CPs;<sup>[19,20]</sup> afterward, the Cu-S signal is intensified in the final pure Cu CPs (Figure 2b). The corresponding scanning electron microscopy (SEM) images of the samples reveal the well-maintained 2D structure of the In CPs substrate with uniform element distributions (Figures S3–S5). Meanwhile, the X-ray photoelectron spectroscopy (XPS) spectra of In 3d are weakened, left-shifted, and eventually disappear with the progressive exchange of  $\text{Cu}^{2+}$  cations (Figure 2c). In contrast, the Cu 2p spectra are gradually intensified and shifted to the right (Figure S6). In this way, the controllable Cu substitution in the structure of the In CPs during the subsequent reconstruction process provides the In atoms with tuned coordination environments, which enables the construction of atomically dispersed In sites in Cu-based nanosheets.

To monitor the structural reconstruction process of Cu-In CPs in neutral media, in situ Raman spectroscopy characterization was conducted. As the pretreatment time increases, the decreased intensity of the Raman vibration peaks ascribed to the stretch of  $\nu(\text{C}=\text{N})$  and  $\nu(\text{C}-\text{N})$  at 1370  $\text{cm}^{-1}$  can be observed (Figures 2d and S7), demonstrating the gradual detachment of the ligand units of the bimetallic coordination polymers. The characteristic peaks at 1347 and 1602  $\text{cm}^{-1}$  are assigned to the D and G bands of carbon black,<sup>[21]</sup> which is introduced to enhance the dispersion and conductivity of the precursors.<sup>[22]</sup> Simultaneously, the typical Cu-S bond is gradually transformed into the characteristic



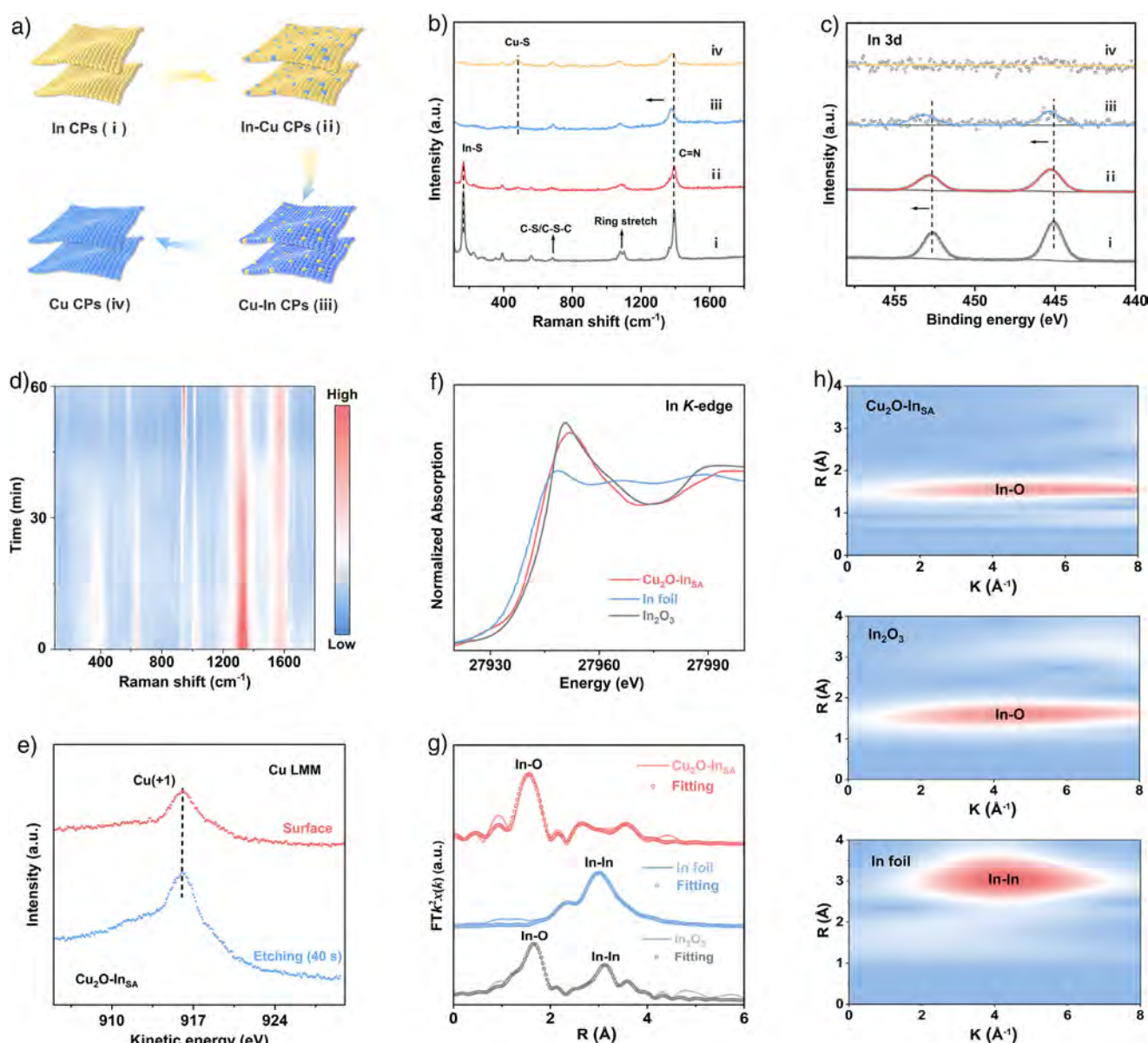
**Figure 1.** Synthesis of  $\text{Cu}_2\text{O-InSA}$  nanosheets. a) Schematic illustration of the synthesis process of  $\text{Cu}_2\text{O-InSA}$ . b) TEM, HAADF-STEM, and the corresponding EDS mapping images of the Cu-In CPs precursor. c) TEM, d) AFM, and e) AC-HAADF-STEM images of  $\text{Cu}_2\text{O-InSA}$ . f) Corresponding intensity profiles. g) HAADF-STEM and the corresponding EDS mapping images of  $\text{Cu}_2\text{O-InSA}$ .

Cu–O bonds at around 163, 427, and  $628\text{ cm}^{-1}$ , which can be ascribed to the formation of  $\text{Cu}_2\text{O}$ .<sup>[23]</sup> Interestingly, the peaks corresponding to the  $\text{NO}_3^-$  ions and  $\text{SO}_4^{2-}$  ions gradually increased, indicating that the reconstructed  $\text{Cu}_2\text{O-InSA}$  structure is beneficial for adsorbing oxoanions, which facilitate the subsequent reduction reaction.<sup>[24]</sup> Moreover, the depth-profiling XPS spectra of the Cu LMM Auger region of the Cu-In CPs sample after pretreatment show one strong signal at around 916 eV (Figure 2e),<sup>[25]</sup> indicating the formation of the  $\text{Cu}_2\text{O}$  phase, in agreement with the in situ Raman results.

X-ray absorption fine structure (XAFS) spectroscopy was performed to unveil the atomic coordination environment of In sites in the reconstructed  $\text{Cu}_2\text{O-InSA}$ . The In *K*-

edge X-ray absorption near-edge structure (XANES) spectra indicate that the absorption edge lies between In foil and  $\text{In}_2\text{O}_3$ , demonstrating the existence of positively charged  $\text{In}^{\delta+}$  ( $0 < \delta < 3$ ) in the  $\text{Cu}_2\text{O}$  (Figure 2f).<sup>[26]</sup> The low-valent feature endows the In SAs with amphiphilicity, owing to the coexistence of vacant p-orbital and lone electron pair, potentially acting as both a Lewis acid site and a Lewis base site during the  $\text{NO}_3^-$ -RR.<sup>[27]</sup> The Fourier transform (FT) extended X-ray adsorption fine structure (EXAFS) spectra and the corresponding fitting results display a major peak at around  $2.09\text{ \AA}$  corresponding to the scattering of In–O bonding and no scattering peaks for In–In metallic bonding is detected (Figure 2g and Table S1), indicating that the In species is in an isolated atomic state in the lattice of  $\text{Cu}_2\text{O}$ .





**Figure 2.** Atomic structural and chemical states analysis. a) Schematic illustration of the gradient cation exchange strategy. Raman spectra (b) and In 3d XPS spectra (c) of In CPs, In-Cu CPs, Cu-In CPs, and Cu CPs. d) Contour plot of in situ Raman spectra for the reconstruction process of Cu-In CPs into Cu<sub>2</sub>O-In<sub>2</sub>O<sub>3</sub>. e) Depth-profiling Cu LMM Auger spectra for Cu<sub>2</sub>O-In<sub>2</sub>O<sub>3</sub>. In K-edge XANES spectra (f) and FT k<sup>3</sup>-weighted EXAFS spectra (g) of Cu<sub>2</sub>O-In<sub>2</sub>O<sub>3</sub> and reference samples. h) EXAFS wavelet transform plots.

nanosheets,<sup>[28]</sup> which is further confirmed by the Wavelet transform plots (Figure 2h).

To further disclose the effect of the structural evolution on the adsorption behavior toward NO<sub>3</sub><sup>-</sup> ions, the in situ Raman spectroscopy over the Cu CPs, In CPs, and In-Cu CPs precursors was performed under identical conditions. For the Cu CPs, the bonds ascribed to Cu<sub>2</sub>O at around 163, 446, and 637 cm<sup>-1</sup> are generated, accompanied by the detachment of the ligand units, demonstrating that the Cu CPs precursor is reconstructed into Cu<sub>2</sub>O (Figure S8). The corresponding depth-profiling XPS spectra of the Cu LMM Auger region confirm the formation of Cu (+1) species (Figure S9). Moreover, the negative shift of Cu 2p<sub>3/2</sub> and Cu 2p<sub>1/2</sub> peaks of Cu<sub>2</sub>O-In<sub>2</sub>O<sub>3</sub> compared to the Cu<sub>2</sub>O counterparts

implies the strong electron interaction between In SAs and Cu<sub>2</sub>O (Figure S10).<sup>[29]</sup> Besides, the reconstruction process of Cu CPs also exhibits gradually enhanced oxoanion adsorption corresponding to the NO<sub>3</sub><sup>-</sup> and SO<sub>4</sub><sup>2-</sup> ions, which is similar to that of Cu-In CPs, suggesting that the derived Cu<sub>2</sub>O and Cu<sub>2</sub>O-In<sub>2</sub>O<sub>3</sub> catalysts engender a larger number of adsorption sites toward NO<sub>3</sub><sup>-</sup> ions. In addition, the In CPs and In-Cu CPs precursors are reconstructed into amorphous InO<sub>x</sub> and Cu-InO<sub>x</sub>, which are corroborated by the corresponding in situ Raman spectroscopy and XPS results (Figures S11 and S12).<sup>[30]</sup> Interestingly, the In-rich CPs show strong adsorption peaks of SO<sub>4</sub><sup>2-</sup> ions before the reconstructions, whereas no obvious Raman peaks of oxoanion are detected over the derived InO<sub>x</sub> and Cu-InO<sub>x</sub>.<sup>[31]</sup> Therefore, the

above findings suggest that the synergetic In/Cu sites with specific chemical environments can potentially modulate the adsorption behaviors and the interfacial concentration of the oxoanions, which will be further explored and verified in detail.

### Electrocatalytic Performance of $\text{NO}_3^-$ -RR

In the following, electrochemical  $\text{NO}_3^-$ -RR measurements were conducted using a three-electrode system in an H-cell at room temperature with a 0.5 M  $\text{Na}_2\text{SO}_4$  solution containing 0.1 M  $\text{KNO}_3$  as electrolyte. The products were quantified by ultraviolet–visible spectroscopy (UV–vis) (Figures S13 and S14).<sup>[32]</sup> The  $\text{Cu}_2\text{O-In}_{\text{SA}}$  sample and the other three counterparts (i.e.,  $\text{InO}_x$ ,  $\text{Cu-InO}_x$ , and  $\text{Cu}_2\text{O}$ ) were evaluated for the performance of  $\text{NO}_3^-$ -RR to  $\text{NH}_3$ . As depicted in the linear sweep voltammetry (LSV) curves (Figure 3a), the current densities of all samples are enhanced after adding the  $\text{NO}_3^-$  ions, suggesting that the  $\text{NO}_3^-$ -RR is relatively active compared to the competitive HER. Notably, the increased current densities of  $\text{Cu}_2\text{O-In}_{\text{SA}}$  and  $\text{Cu}_2\text{O}$  are more pronounced than those of the  $\text{InO}_x$  and  $\text{Cu-InO}_x$ , suggesting that affluent Cu sites are auspicious to the  $\text{NO}_3^-$ -RR. Additionally, the  $\text{Cu}_2\text{O-In}_{\text{SA}}$  electrocatalyst exhibits higher current density than  $\text{Cu}_2\text{O}$  in the presence or absence of  $\text{NO}_3^-$  ions, demonstrating that the In SAs can enhance the electrochemical  $\text{NO}_3^-$ -RR capability of  $\text{Cu}_2\text{O}$ . As shown in Figure 3b,c, the  $\text{Cu}_2\text{O-In}_{\text{SA}}$  sample also exhibits superior Faradaic efficiency (FE) and yield rate for electrocatalytic conversion of  $\text{NO}_3^-$  to  $\text{NH}_3$ . Specifically, at  $-0.8$  V versus RHE,  $\text{Cu}_2\text{O-In}_{\text{SA}}$  presents a  $\text{FE}_{\text{NH}_3}$  of 99.36% with an  $\text{NH}_3$  yield rate of  $23.37 \text{ mg h}^{-1} \text{ mg}_{\text{cat}}^{-1}$ . Besides, an impressive  $\text{NH}_3$  yield rate of  $29.02 \text{ mg h}^{-1} \text{ mg}_{\text{cat}}^{-1}$  can also be achieved at  $-0.9$  V versus RHE, with a  $\text{FE}_{\text{NH}_3}$  of 95.60%. In addition, the  $\text{InO}_x$  and  $\text{Cu-InO}_x$  counterparts exhibit much higher  $\text{NO}_2^-$  FE than the  $\text{Cu}_2\text{O}$  and  $\text{Cu}_2\text{O-In}_{\text{SA}}$  (Figure S15), suggesting that the affluent Cu sites are favorable for triggering water dissociation to promote the  $^*\text{H}$  formation for hydrogenation of nitrogen-containing intermediates to  $\text{NH}_3$ .<sup>[33]</sup> To investigate the effect of concentration and the existing form of In species on the catalytic performance, the physically mixed  $\text{InO}_x\text{-Cu}_2\text{O}$  sample and other In SAs-related catalysts were further investigated (Figures S16–S20). The residual and excessive  $\text{InO}_x$  both deteriorate the activity, indicating that the In SA catalyst with a built-in optimized electronic structure is favorable for the  $\text{NH}_3$  generation. Moreover, the double-layer capacitance ( $C_{\text{dl}}$ ) value of  $\text{Cu}_2\text{O-In}_{\text{SA}}$  is measured to be  $7.39 \text{ mF cm}^{-2}$ , which is superior to that of  $\text{Cu}_2\text{O}$  ( $4.18 \text{ mF cm}^{-2}$ ),  $\text{Cu-InO}_x$  ( $1.99 \text{ mF cm}^{-2}$ ), and  $\text{InO}_x$  ( $1.20 \text{ mF cm}^{-2}$ ) (Figures S21 and S22), indicating high electrochemically surface areas and rich catalytically active sites on  $\text{Cu}_2\text{O-In}_{\text{SA}}$ .

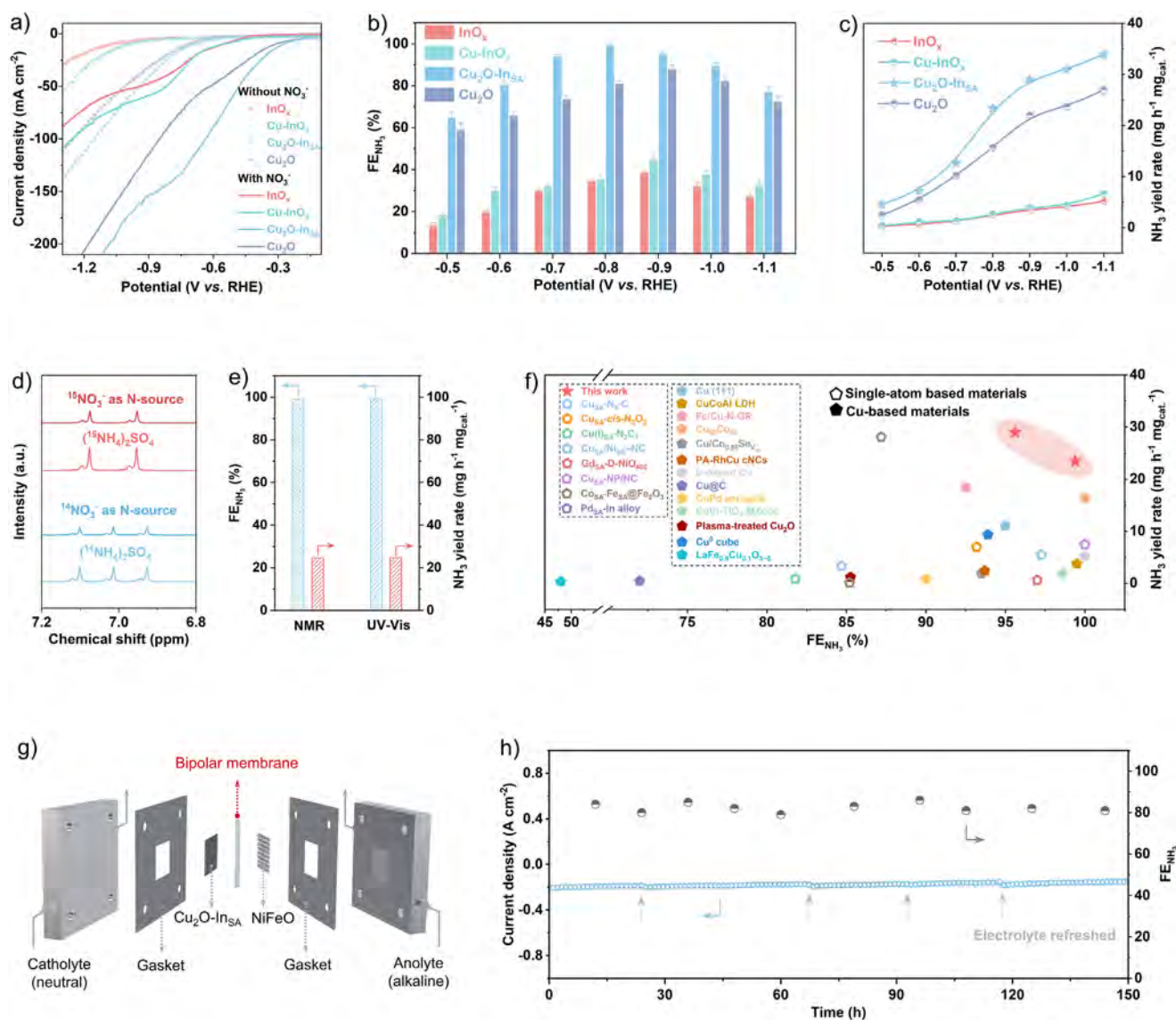
Furthermore, the  $^{15}\text{N}$  isotope labeling experiments were implemented to certify the sources of the  $\text{NH}_3$  product (Figure 3d). The  $^1\text{H}$  nuclear magnetic resonance ( $^1\text{H}$  NMR) spectrum of the product displays triple peaks when  $\text{K}^{14}\text{NO}_3$  is adopted as a reactant. The  $^1\text{H}$  NMR spectrum exhibits double peaks of  $^{15}\text{NH}_4^+$  in the case of  $\text{K}^{15}\text{NO}_3$  as the

feedstock, confirming that the detected  $\text{NH}_3$  product comes from the electrochemical  $\text{NO}_3^-$ -RR process rather than the contamination. Meanwhile, based on the NMR quantification method of  $^{15}\text{NH}_3$  (Figure S23), the value of generated  $^{15}\text{NH}_3$  from electrolyzing  $\text{K}^{15}\text{NO}_3$  at  $-0.8$  V versus RHE is very close to the result measured by the UV–vis method (Figure 3e), corroborating the reliability of the  $\text{NH}_3$  production measurements. To highlight the durability of  $\text{Cu}_2\text{O-In}_{\text{SA}}$  for the  $\text{NO}_3^-$ -RR, a consecutive cycling test was performed (Figure S24). The  $\text{Cu}_2\text{O-In}_{\text{SA}}$  can maintain a high  $\text{FE}_{\text{NH}_3}$  and  $\text{NH}_3$  yield rate with only slight fluctuations after 30 consecutive cycles. XPS analysis reveals that the signals in the In 3d region exhibit no discernible shift, indicating the stable chemical state of In SAs during long-term operation, without the formation of metallic In aggregation (Figure S25). In those regards, the overall performance of  $\text{Cu}_2\text{O-In}_{\text{SA}}$  ranks in the top tier among the recently reported state-of-the-art electrocatalysts for  $\text{NO}_3^-$ -RR (Figure 3f; Tables S2 and S3).

To further demonstrate the large-scale application potential of the  $\text{Cu}_2\text{O-In}_{\text{SA}}$  electrocatalyst for ammonia synthesis, we assembled the  $\text{Cu}_2\text{O-In}_{\text{SA}}$  electrode within a high-throughput bipolar membrane electrode assembly (MEA) reactor (Figure 3g). Specifically, we spray-coated  $\text{Cu}_2\text{O-In}_{\text{SA}}$  on carbon fiber paper (CFP) as a cathode. We selected a commercial  $\text{NiFeO/Ni}$  fiber as an anode. The potentiostatic experiment was conducted with 0.5 M  $\text{Na}_2\text{SO}_4$  and 0.1 M  $\text{KNO}_3$  as the catholyte and 1 M  $\text{KOH}$  as the anolyte. The utilization of a bipolar membrane can prevent the crossover of produced  $\text{NH}_3$  to the anode from being re-oxidized again under high current density (Figure S26).<sup>[34]</sup> It is found that the reactor can maintain a steady current density-time profile at  $200 \text{ mA cm}^{-2}$  for 150 h with a high  $\text{NH}_3$  faradaic efficiency of  $\sim 83\%$  (Figure 3h). Additionally, a photovoltaic-electrolysis system for ammonia synthesis was also constructed through the integration of a two-electrode setup and a commercial silicon solar cell (Figure S27a). The  $\text{Cu}_2\text{O-In}_{\text{SA}}$  cathode proceeds electrocatalytic  $\text{NO}_3^-$ -RR without observation of hydrogen bubbles. In contrast, many oxygen bubbles emerge on the surface of the coupled  $\text{NiFeO/Ni}$  fiber anode (Figure S27b,c). Moreover, as indicated by the indophenol blue method, ammonia production increases with prolonged reaction time (Figure S27d). Therefore, the results demonstrate that the  $\text{Cu}_2\text{O-In}_{\text{SA}}$  electrode possesses great potential for industrial ammonia synthesis applications.

### Mechanism Exploration

To study the real active species in the reductive process, AC-HAADF-STEM analysis was conducted on the  $\text{Cu}_2\text{O-In}_{\text{SA}}$  after the  $\text{NO}_3^-$ -RR process at  $-0.8$  V versus RHE. The corresponding image reveals a typical lattice fringe of 0.21 nm, which can be ascribed to the (111) plane of metallic Cu (Figure 4a).<sup>[35,36]</sup> Meanwhile, the brighter dots marked by red circles demonstrate the well-maintained dispersion of In SAs. Moreover, the TEM and elemental mapping images reveal a well-sustained 2D nanosheet morphology and a well-dispersed distribution of Cu and In elements (Figure S28). The X-ray diffraction (XRD) pattern of  $\text{Cu}_2\text{O-In}_{\text{SA}}$  electrode



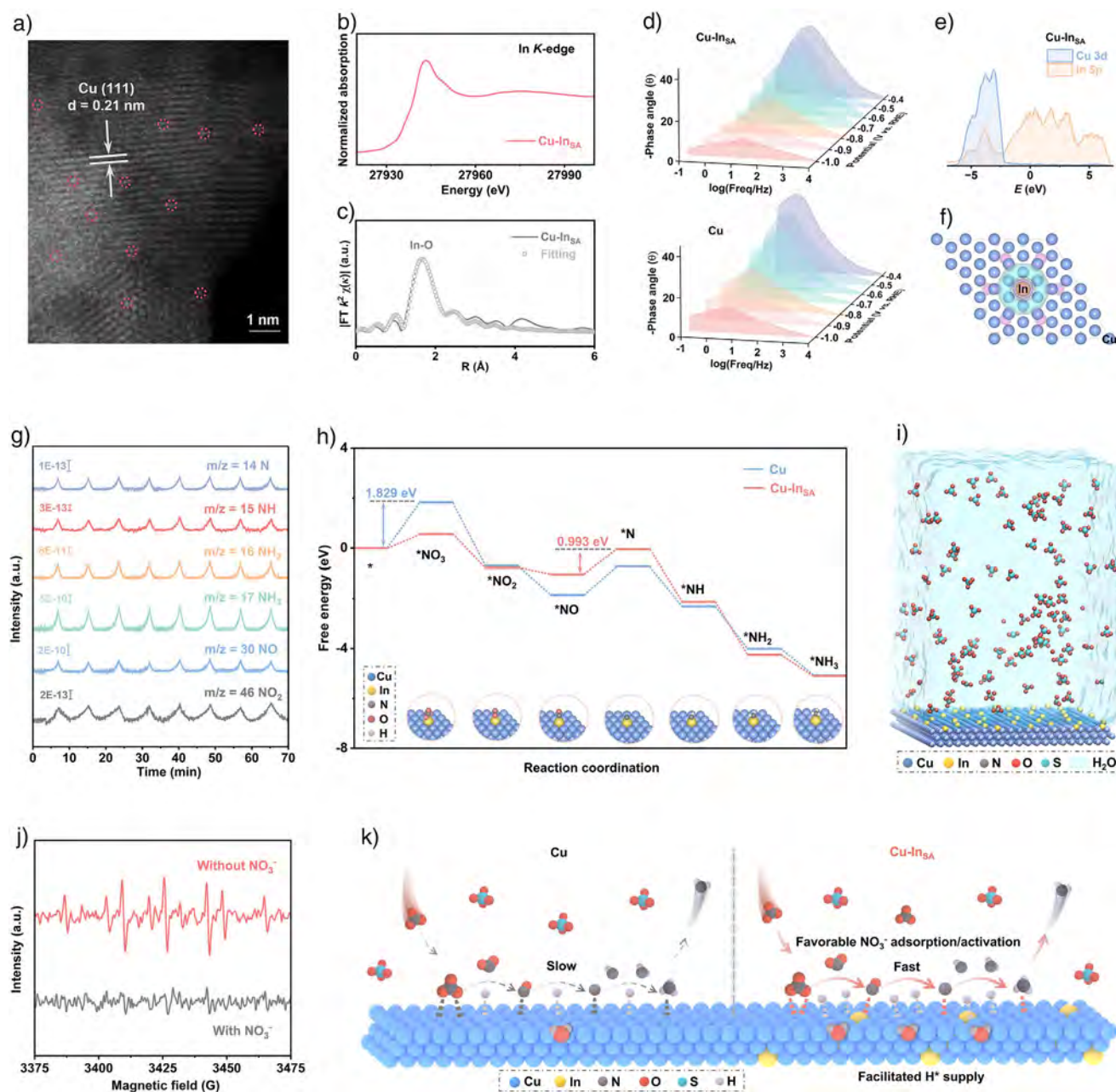
**Figure 3.** Electrocatalytic performance of  $\text{NO}_3^-$  RR. a) LSV curves in an electrolyte of 0.5 M  $\text{Na}_2\text{SO}_4$  with and without 0.1 M  $\text{KNO}_3$ . b) Faradaic efficiencies for  $\text{NH}_3$  ( $\text{FE}_{\text{NH}_3}$ ). c)  $\text{NH}_3$  yield rates over the different samples (the error bars represent the standard deviations determined from three independent experimental tests). d)  $^1\text{H}$  NMR spectra of electrolytes after the reaction at  $-0.8$  V (versus RHE) using 0.1 M  $^{15}\text{NO}_3^-$  and  $^{14}\text{NO}_3^-$  as nitrogen sources. e) The calculated  $\text{FE}_{\text{NH}_3}$  and  $\text{NH}_3$  yield rates using the NMR and UV-vis methods. f) Comparison of the  $\text{NO}_3^-$  RR performance of the  $\text{Cu}_2\text{O-InSA}$  with single-atom and Cu-based materials. g) Schematic illustration of  $\text{NH}_3$  electrosynthesis in a bipolar membrane electrode assembly reactor. h) Time-dependent current density and  $\text{FE}_{\text{NH}_3}$  over the  $\text{Cu}_2\text{O-InSA}$  electrode.

after the  $\text{NO}_3^-$  RR displays the typical diffraction peak at  $43^\circ$  corresponding to the (111) plane of the metallic Cu phase (JCPDS no.04-0836) (Figure S29).<sup>[37]</sup> Therefore, these results reveal that the  $\text{Cu}_2\text{O}$  substrate is self-adaptively reconstructed into metallic Cu nanosheets during the  $\text{NO}_3^-$  RR process (denoted as  $\text{Cu-InSA}$ ) (Figure S30). Likewise, the pure  $\text{Cu}_2\text{O}$  counterpart also undergoes a reconstructed process into metallic Cu, as supported by the results of XRD and HRETM (Figures S29 and S31). Furthermore, the local fine structure of the In site in the  $\text{Cu-InSA}$  was investigated using XAFS measurement. The In *K*-edge XANES spectrum of  $\text{Cu-InSA}$  is exhibited in Figure 4b. The FT-EXAFS spectrum and corresponding fitting results display no In-In coordination bond at around 3.2 Å, suggesting that the In atoms maintain an

atomically dispersed state without aggregation in the  $\text{Cu-InSA}$  after the  $\text{NO}_3^-$  RR process (Figure 4c and Table S4).<sup>[38,39]</sup>

To unveil the role of In SAs on the interfacial selective adsorption behavior of ions during the  $\text{NO}_3^-$  RR process, the in situ Raman spectroscopy was conducted under different applied potentials (Figure S32a). The Raman characteristic peaks of the  $\text{Cu}_2\text{O}$  matrix and the adsorption behaviors toward  $\text{NO}_3^-$  ions are well preserved from open circuit potential (OCP) to  $-0.7$  V versus RHE. After further increasing the applied negative potential, the relative intensity ratio of vibration peaks associated with the adsorption of  $\text{SO}_4^{2-}$  and  $\text{NO}_3^-$  ions exhibits apparent inversion (Figure S32b). The affinity toward  $\text{SO}_4^{2-}$  generally remains stable for the reported Cu-based catalysts as the potential becomes





**Figure 4.** Insights into the  $\text{NO}_3^-$  RR to  $\text{NH}_3$  mechanism. a) AC-HADDF-STEM of  $\text{Cu}_2\text{O-InSA}$  after  $\text{NO}_3^-$  RR (i.e.,  $\text{Cu-InSA}$ ). b) The normalized XANES at the In K-edge of  $\text{Cu-InSA}$ . c) FT-EXAFS fitting curve of  $\text{Cu-InSA}$ . d) Bode phase plots of  $\text{Cu-InSA}$  and Cu at varied potentials. e) PDOS of  $\text{Cu-InSA}$ . f) Charge density differences of  $\text{Cu-InSA}$  (blue: electron accumulation; purple: electron depletion). g) Online DEMS measurements of  $\text{NO}_3^-$  RR over  $\text{Cu-InSA}$ . h) Calculated free energy diagrams for  $\text{NO}_3^-$  RR. i) MD simulation of  $\text{Cu-InSA}$  in 0.5 M  $\text{Na}_2\text{SO}_4$  solution containing 0.1 M  $\text{KNO}_3$ . j) DMPO spin-trapping EPR spectra for  $\cdot\text{H}$  detection toward  $\text{Cu-InSA}$  with and without  $\text{NO}_3^-$ . k) Schematic diagram of  $\text{NO}_3^-$  RR to  $\text{NH}_3$  process on  $\text{Cu-InSA}$  and Cu.

increasingly negative.<sup>[40]</sup> Here, the results demonstrate a gradually pronounced  $\text{NO}_3^-$  enrichment effect with  $\text{SO}_4^{2-}$  repelling at the interface of electrode-electrolyte during the  $\text{NO}_3^-$  RR process.<sup>[16]</sup> Meanwhile, the Raman peaks belonging to  $\text{Cu}_2\text{O}$  disappear owing to its reduction to metallic Cu, which is consistent with the above observations. The new Raman peak, appropriately around  $1505\text{ cm}^{-1}$ , assigned to -NH intermediate, appears with increased potential, verifying the hydrogenation process.<sup>[41]</sup> Moreover, it can be deduced

that, from OCP to  $-0.7\text{ V}$ , the incompletely reconstructed  $\text{Cu}_2\text{O-InSA}$  with residual electronegative oxygen atoms could induce a stronger electric field, which is beneficial for the Coulomb attraction of  $\text{SO}_4^{2-}$  with high localized charge density. After self-adaptively reconstructing into  $\text{Cu-InSA}$ , the In SA features a vacant p-orbital (Lewis acidity) to adsorb/polarize  $\text{NO}_3^-$  and acts as a lone-pair/electron reservoir coupled to metallic Cu for electric field-guided proton-coupled electron transfer, yielding higher and more

stable  $\text{NH}_3$  selectivity. For comparison, an analogous trend of  $\text{Cu}_2\text{O}$  phase reduced into metallic Cu phase is observed over the  $\text{Cu}_2\text{O}$  counterpart (Figure S33a). However, as negative potentials are increasingly applied, the intensity ratio of vibration peaks related to the adsorption of  $\text{SO}_4^{2-}$  and  $\text{NO}_3^-$  displays slight changes (Figure S33b), implying that the introduction of the low-valent In sites in the Cu matrix with vacant *p*-orbital and lone electron pair can effectively tune the dielectric microenvironment, thus enhancing the supplement of interfacial  $\text{NO}_3^-$  ions.

The operando electrochemical impedance spectra (EIS) were applied to explore the interfacial reaction kinetics of  $\text{NO}_3^-$  RR to  $\text{NH}_3$  (Figure 4d). As a result, the Bode plots of Cu–In<sub>SA</sub> with  $\text{NO}_3^-$  ions show smaller phase angles at low frequency and more significant declined peak intensity compared to those of Cu (derived from the  $\text{Cu}_2\text{O}$  counterparts), suggesting that the introduction of In SAs can accelerate the  $^*\text{H}$  consumption with  $\text{NO}_3^-$  ions and enhance the interfacial charge transfer with intensified  $\text{NO}_3^-$  adsorption during  $\text{NO}_3^-$  RR.<sup>[15]</sup> In the following, the density functional theory (DFT) calculations are conducted to investigate the underlying  $\text{NO}_3^-$  RR mechanism. The projected density of states (PDOS) of Cu–In<sub>SA</sub> reveals a strong orbital overlap among Cu 3d and In 5p orbitals (Figure 4e).<sup>[42]</sup> More importantly, as compared with pure Cu (Figure S34), the introduction of In SAs induces much higher occupation near the Fermi level ( $E_f$ ) for Cu–In<sub>SA</sub>, which demonstrates a promoted electron transfer in Cu–In<sub>SA</sub> by the strong p-d orbital coupling,<sup>[43]</sup> aligning with the operando EIS results. As depicted by the differential charge density of  $\text{Cu}_2\text{O}$ –In<sub>SA</sub> with geometric optimization (Figure 4f), more electrons are delocalized around In SAs, illustrating that the In SAs regulate the local electronic distribution of the Cu matrix to tune the electron transportation and intermediate adsorption.<sup>[44]</sup> Moreover, the quantitative assessment using Bader charge discloses that the In center donates 0.34 electrons to the adjacent Cu atoms, forming a unique polarized In–Cu pair.

To gain a more comprehensive understanding of the reaction mechanism, online differential electrochemical mass spectrometry (DEMS) was employed to monitor key intermediates and products (Figures 4g and S35).<sup>[45]</sup> It reveals the *m/z* signals of 46, 30, 17, 16, 15, and 14, corresponding to  $^*\text{NO}_2$ ,  $^*\text{NO}$ ,  $^*\text{NH}_3$ ,  $^*\text{NH}_2$ ,  $^*\text{NH}$ , and  $^*\text{N}$ , respectively.<sup>[46]</sup> Therefore, the  $\text{NO}_3^-$  RR pathway can be proposed: the  $\text{NO}_3^-$  is initially adsorbed on the surface ( $^*\text{NO}_3$ ), subsequently reduced to  $^*\text{NO}_2$  and  $^*\text{NO}$ ; then, the  $^*\text{NO}$  is protonated into  $^*\text{N}$ ,  $^*\text{NH}_2$ , and  $\text{NH}_3$  as the final product. Based on the DEMS results, we calculated the free energy of individual intermediates on Cu–In<sub>SA</sub> and Cu (Figure 4h). The conversion from  $^*\text{NO}$  to  $^*\text{N}$  is identified as the rate-determining step (RDS) for Cu–In<sub>SA</sub>, characterized by a lower energy barrier of 0.993 eV. At the same time, the RDS for Cu is the activation of adsorbed  $\text{NO}_3^-$  with a larger energy barrier of 1.829 eV, highlighting the critical role of In SAs in promoting the kinetics of the  $\text{NO}_3^-$  RR process.<sup>[47]</sup> Moreover, it is also indicated that the introduction of In SAs can tune the electronic structure of Cu–In<sub>SA</sub> for favorable  $\text{NO}_3^-$  adsorption and activation.

Based on the above experimental and theoretical findings, molecular dynamics (MD) simulations were conducted to

quantitatively analyze the dynamic behavior of  $\text{NO}_3^-$  at the interface of a Cu–In<sub>SA</sub> electrode with the electrolyte in a neutral  $\text{Na}_2\text{SO}_4$  solution (Figure 4i).<sup>[48]</sup> The pure Cu electrode was constructed as a comparison (Figure S36). The resulting radial distribution function (RDF) curves show a stronger interaction between Cu–In<sub>SA</sub> and  $\text{NO}_3^-$  than that between Cu (Figure S37). The concentrations of  $\text{NO}_3^-$  distributed on the Cu–In<sub>SA</sub> surface are higher than those on the Cu surface. Meanwhile, the Cu–In<sub>SA</sub> can retard  $\text{SO}_4^{2-}$  ions by creating a longer distance of  $\text{SO}_4^{2-}$  accumulation from the surface as compared to that from the Cu surface. Furthermore, the Bader charge analysis reveals that  $\text{NO}_3^-$  gains more electronic charge from the Cu–In<sub>SA</sub> surface than  $\text{SO}_4^{2-}$  (Figure S38), indicating a stronger interaction of  $\text{NO}_3^-$  compared to  $\text{SO}_4^{2-}$ .<sup>[49]</sup> Those results further confirm the critical role of the In sites in the Cu-based matrix in selectively adsorbing  $\text{NO}_3^-$  by repelling  $\text{SO}_4^{2-}$  at the electrocatalyst/electrolyte interface, thereby endowing the Cu–In<sub>SA</sub> to act as a solid-state buffer that tunes the interfacial ion equilibrium to maintain the high kinetics of the  $\text{NO}_3^-$  RR.

Afterward, the water adsorption and dissociation processes are calculated, owing to the supply of the necessary  $^*\text{H}$  for  $\text{NO}_3^-$  RR (Figure S39).<sup>[6]</sup> As a result, the Gibbs free energy change for water dissociation on Cu (1.42 eV) is higher than that of Cu–In<sub>SA</sub> (1.34 eV), implying that In SAs can optimize the Cu matrix to improve water dissociation for more  $^*\text{H}$  supply in the subsequent hydrogenation processes.<sup>[50]</sup> To obtain further evidence for the behaviors of active hydrogen (i.e.,  $^*\text{H}$ ), we conducted electron paramagnetic resonance (EPR) measurements. The EPR spectra were collected in 0.5 M  $\text{Na}_2\text{SO}_4$  with and without  $\text{NO}_3^-$ , utilizing 5,5-dimethyl-1-pyrroline-*N*-oxide (DMPO) as the agent for trapping hydrogen radicals. As depicted in Figure 4j, after adding  $\text{NO}_3^-$  to the solution, the peak intensity significantly decreases, suggesting that the produced hydrogen radicals are efficiently consumed with nitrogen-containing intermediates in the hydrogenation reactions.<sup>[51]</sup> In this way, due to the strong *d*–*p* orbital overlaps between Cu sites and In SAs, the self-adaptive Cu–In<sub>SA</sub> system becomes favorable for  $\text{NO}_3^-$  adsorption and activation, lowers the energy barrier of RDS, promotes the water dissociation to facilitate  $^*\text{H}$  supply for supporting  $\text{NO}_3^-$  RR activity (Figure 4k).

## Conclusions

In summary, high-performance ultrathin In SAs-embedded Cu-based nanosheets for the  $\text{NO}_3^-$  RR to  $\text{NH}_3$  were controllably synthesized via a facile topological electrochemical reconstruction strategy. Combining operando experiments, theoretical calculations, and MD simulations, it is found that the low-valent In SAs decorated the self-adaptive Cu-based nanosheet enable optimal local electronic distribution by strong p-d orbital couplings, thus facilitating electron transfer, accelerating  $^*\text{H}$  supply, and lowering the RDS energy barrier. Meanwhile, the In SAs endow the Cu-based species as a solid-state buffer by strengthening the interfacial adsorption of the  $\text{NO}_3^-$  ion, while repelling  $\text{SO}_4^{2-}$  supporting ions when the  $\text{NO}_3^-$  RR proceeds continuously, thereby consuming  $\text{NO}_3^-$



ions and sustaining a high  $\text{NO}_3^-$ -RR kinetics. Therefore, under neutral conditions, the electrocatalyst can achieve a maximum  $\text{FE}_{\text{NH}_3}$  of 99.36%, a high  $\text{NH}_3$  yield rate of  $29.02 \text{ mg h}^{-1} \text{ mg}_{\text{cat.}}^{-1}$ , and excellent electrocatalytic durability over 30 cycles, which ranks the top tier among the recently reported state-of-the-art electrocatalysts for  $\text{NO}_3^-$ -RR. Besides, the as-designed electrode can be operated stably in an MEA electrolyzer at  $200 \text{ mA cm}^{-2}$  for 150 h with an average  $\text{FE}_{\text{NH}_3}$  of  $\sim 83\%$ . It can also be effectively driven by a solar cell under sunlight irradiation for  $\text{NH}_3$  synthesis, demonstrating promising industrial applications.

## Acknowledgements

This work was financially supported by the City University of Hong Kong (project no. 9229138, 9231502, and 9231539), the Guangdong Provincial Science and Technology Plan Project (no. 2025A0505080006), and the Guangdong Provincial Basic and Applied Basic Research Project (no. 2024B1515120005).

## Conflict of Interests

The authors declare no conflict of interest.

## Data Availability Statement

The data that support the findings of this study are available from the corresponding author upon reasonable request.

**Keywords:** Hydrogen radical • Nitrate reduction reaction • Selective adsorption • Single atom • Solid-state buffering

- [1] S. Han, H. Li, T. Li, F. Chen, R. Yang, Y. Yu, B. Zhang, *Nat. Catal.* **2023**, 6, 402–414.
- [2] J. G. Chen, R. M. Crooks, L. C. Seefeldt, K. L. Bren, R. M. Bullock, M. Y. Darensbourg, P. L. Holland, B. Hoffman, M. J. Janik, A. K. Jones, M. G. Kanatzidis, P. King, K. M. Lancaster, S. V. Lyman, P. Pfromm, W. F. Schneider, R. R. Schrock, *Science* **2018**, 360, eaar6611.
- [3] H. Xu, Y. Ma, J. Chen, W.-X. Zhang, J. Yang, *Chem. Soc. Rev.* **2022**, 51, 2710–2758.
- [4] B. Zhou, L. Yu, W. Zhang, X. Liu, H. Zhang, J. Cheng, Z. Chen, H. Zhang, M. Li, Y. Shi, F. Jia, Y. Huang, L. Zhang, Z. Ai, *Angew. Chem. Int. Ed.* **2024**, 63, e202406046.
- [5] W. Wen, S. Fang, Y. Zhou, Y. Zhao, P. Li, X.-Y. Yu, *Angew. Chem. Int. Ed.* **2024**, 63, e202408382.
- [6] X. Zhang, X. Liu, Z.-F. Huang, L. Gan, S. Zhang, R. Jia, M. Ajmal, L. Pan, C. Shi, X. Zhang, G. Yang, J.-J. Zou, *Energy Environ. Sci.* **2024**, 17, 6717–6727.
- [7] L. Bai, F. Franco, J. Timoshenko, C. Rettenmaier, F. Scholten, H. S. Jeon, A. Yoon, M. Rüscher, A. Herzog, F. T. Haase, S. Kühl, S. W. Chee, A. Bergmann, R. C. Beatriz, *J. Am. Chem. Soc.* **2024**, 146, 9665–9678.
- [8] R. Daiyan, T. Tran-Phu, P. Kumar, K. Iputera, Z. Tong, J. Leverett, M. H. A. Khan, A. Asghar Esmailpour, A. Jalili, M. Lim, A. Tricoli, R.-S. Liu, X. Lu, E. Lovell, R. Amal, *Energy Environ. Sci.* **2021**, 14, 3588–3598.
- [9] J.-Y. Fang, Q.-Z. Zheng, Y.-Y. Lou, K.-M. Zhao, S.-N. Hu, G. Li, O. Akdim, X.-Y. Huang, S.-G. Sun, *Nat. Commun.* **2022**, 13, 7899.
- [10] Y. Wang, A. Xu, Z. Wang, L. Huang, J. Li, F. Li, J. Wicks, M. Luo, D.-H. Nam, C.-S. Tan, Y. Ding, J. Wu, Y. Lum, C.-T. Dinh, D. Sinton, G. Zheng, E. H. Sargent, *J. Am. Chem. Soc.* **2020**, 142, 5702–5708.
- [11] Q. Hu, S. Qi, Q. Huo, Y. Zhao, J. Sun, X. Chen, M. Lv, W. Zhou, C. Feng, X. Chai, H. Yang, C. He, *J. Am. Chem. Soc.* **2024**, 146, 2967–2976.
- [12] S. Zhang, M. Li, J. Li, Q. Song, X. Liu, *Proc. Natl. Acad. Sci. USA* **2022**, 119, e2115504119.
- [13] C. Ma, H. Zhang, J. Xia, X. Zhu, K. Qu, F. Feng, S. Han, C. He, X. Ma, G. Lin, W. Cao, X. Meng, L. Zhu, Y. Yu, A.-L. Wang, Q. Lu, *J. Am. Chem. Soc.* **2024**, 146, 20069–20079.
- [14] F.-Y. Chen, A. Elgazzar, S. Pecaut, C. Qiu, Y. Feng, S. Ashokkumar, Z. Yu, C. Sellers, S. Hao, P. Zhu, H. Wang, *Nat. Catal.* **2024**, 7, 1032–1043.
- [15] S. Liang, X. Teng, H. Xu, L. Chen, J. Shi, *Angew. Chem. Int. Ed.* **2024**, 63, e202400206.
- [16] J. Fan, L. K. Arrazola, J. Du, H. Xu, S. Fang, Y. Liu, Z. Wu, J.-H. Kim, X. Wu, *Environ. Sci. Technol.* **2024**, 58, 12823–12845.
- [17] S. H. Li, S. Hu, H. Liu, J. Liu, X. Kang, S. Ge, Z. Zhang, Q. Yu, B. Liu, *ACS Nano* **2023**, 17, 9338–9346.
- [18] Y. Zhu, J. Wang, T. Koketsu, M. Kroschel, J.-M. Chen, S.-Y. Hsu, G. Henkelman, Z. Hu, P. Strasser, J. Ma, *Nat. Commun.* **2022**, 13, 7754.
- [19] Y. Wu, C. Liu, C. Wang, Y. Yu, Y. Shi, B. Zhang, *Nat. Commun.* **2021**, 12, 3881.
- [20] E. Rudigier, B. Barcones, I. Luck, T. Jawhari-Colin, A. Pérez-Rodríguez, R. Scheer, *J. Appl. Phys.* **2004**, 95, 5153–5158.
- [21] Q. Quan, Y. Zhang, S. Li, S. Yip, W. Wang, P. Xie, D. Chen, W. Wang, D. Yin, Y. Li, B. Liu, J. C. Ho, *ACS Nano* **2024**, 18, 1204–1213.
- [22] Q. Quan, Z. Lai, Y. Bao, X. Bu, Y. Meng, W. Wang, T. Takahashi, T. Hosomi, K. Nagashima, T. Yanagida, C. Liu, J. Lu, J. C. Ho, *Small* **2021**, 17, 2006860.
- [23] M. He, Y. Wu, R. Li, Y. Wang, C. Liu, B. Zhang, *Nat. Commun.* **2023**, 14, 5088.
- [24] A. Kumar, J. Lee, M. G. Kim, B. Debnath, X. Liu, Y. Hwang, Y. Wang, X. Shao, A. R. Jadhav, Y. Liu, H. Tüysüz, H. Lee, *ACS Nano* **2022**, 16, 15297–15309.
- [25] W. Niu, T. Moehl, P. Adams, X. Zhang, R. Lefèvre, A. M. Cruz, P. Zeng, K. Kunze, W. Yang, S. D. Tilley, *Energy Environ. Sci.* **2022**, 15, 2002–2010.
- [26] E. Zhang, L. Tao, J. An, J. Zhang, L. Meng, X. Zheng, Y. Wang, N. Li, S. Du, J. Zhang, D. Wang, Y. Li, *Angew. Chem. Int. Ed.* **2022**, 61, e202117347.
- [27] J. A. J. Pardoe, A. J. Downs, *Chem. Rev.* **2007**, 107, 2–45.
- [28] Y. Guo, Z. Jin, J. Lu, L. Wei, W. Wang, Y. Huang, A. Wang, *Energy Environ. Sci.* **2023**, 16, 5274–5283.
- [29] K. Zhang, B. Li, F. Guo, N. Graham, W. He, W. Yu, *Angew. Chem. Int. Ed.* **2024**, 63, e202411796.
- [30] J. Zhang, R. Yin, Q. Shao, T. Zhu, X. Huang, *Angew. Chem. Int. Ed.* **2019**, 58, 5609–5613.
- [31] W. W. Rudolph, D. Fischer, M. R. Tomney, C. C. Pye, *Phys. Chem. Chem. Phys.* **2004**, 6, 5145–5155.
- [32] G.-F. Chen, Y. Yuan, H. Jiang, S.-Y. Ren, L.-X. Ding, L. Ma, T. Wu, J. Lu, H. Wang, *Nat. Energy* **2020**, 5, 605–613.
- [33] Y. Wang, F. Hao, H. Xu, M. Sun, X. Wang, Y. Xiong, J. Zhou, F. Liu, Y. Hu, Y. Ma, X. Meng, L. Guo, C. Wang, M. Shao, G. Wang, J. Wang, P. Lu, J. Yin, J. Wang, W. Niu, C. Ye, Q. Zhang, S. Xi, B. Huang, M. Shao, Z. Fan, *Angew. Chem. Int. Ed.* **2025**, 64, e202508617.
- [34] Z. Xu, L. Wan, Y. Liao, M. Pang, Q. Xu, P. Wang, B. Wang, *Nat. Commun.* **2023**, 14, 1619.

- [35] B. Zhang, J. Zhang, M. Hua, Q. Wan, Z. Su, X. Tan, L. Liu, F. Zhang, G. Chen, D. Tan, X. Cheng, B. Han, L. Zheng, G. Mo, *J. Am. Chem. Soc.* **2020**, *142*, 13606–13613.
- [36] W. Luc, X. Fu, J. Shi, J.-J. Lv, M. Jouny, B. H. Ko, Y. Xu, Q. Tu, X. Hu, J. Wu, Q. Yue, Y. Liu, F. Jiao, Y. Kang, *Nat. Catal.* **2019**, *2*, 423–430.
- [37] Y. Pan, Y. Li, C.-L. Dong, Y.-C. Huang, J. Wu, J. Shi, Y. Lu, M. Yang, S. Wang, Y. Zou, *Chem* **2023**, *9*, 963–977.
- [38] M. Li, K. Duanmu, C. Wan, T. Cheng, L. Zhang, S. Dai, W. Chen, Z. Zhao, P. Li, H. Fei, Y. Zhu, R. Yu, J. Luo, K. Zang, Z. Lin, M. Ding, J. Huang, H. Sun, J. Guo, X. Pan, W. A. Goddard, P. Sautet, Y. Huang, X. Duan, *Nat. Catal.* **2019**, *2*, 495–503.
- [39] M. Wang, M. Fang, Y. Liu, C. Chen, Y. Zhang, S. Jia, H. Wu, M. He, B. Han, *J. Am. Chem. Soc.* **2025**, *147*, 16450–16458.
- [40] R. Zhao, Q. Yan, L. Yu, T. Yan, X. Zhu, Z. Zhao, L. Liu, J. Xi, *Adv. Mater.* **2023**, *35*, 2306633.
- [41] S. Liu, T. Qian, M. Wang, H. Ji, X. Shen, C. Wang, C. Yan, *Nat. Catal.* **2021**, *4*, 322–331.
- [42] Z. Zeng, L. Y. Gan, H. Bin Yang, X. Su, J. Gao, W. Liu, H. Matsumoto, J. Gong, J. Zhang, W. Cai, Z. Zhang, Y. Yan, B. Liu, P. Chen, *Nat. Commun.* **2021**, *12*, 4088.
- [43] Q. Quan, Y. Zhang, H. Li, W. Wang, P. Xie, D. Chen, W. Wang, Y. Meng, D. Yin, Y. Li, D. Song, L. Chen, S. Li, C. Yang, T. Yanagida, C.-Y. Wong, S. Yip, J. C. Ho, *Nat. Commun.* **2025**, *16*, 2908.
- [44] G. Zhang, F. Wang, K. Chen, J. Kang, K. Chu, *Adv. Funct. Mater.* **2024**, *34*, 2305372.
- [45] J. Dai, Y. Tong, L. Zhao, Z. Hu, C.-T. Chen, C.-Y. Kuo, G. Zhan, J. Wang, X. Zou, Q. Zheng, W. Hou, R. Wang, K. Wang, R. Zhao, X.-K. Gu, Y. Yao, L. Zhang, *Nat. Commun.* **2024**, *15*, 88.
- [46] H. Yin, Z. Chen, S. Xiong, J. Chen, C. Wang, R. Wang, Y. Kuwahara, J. Luo, H. Yamashita, Y. Peng, J. Li, *Chem Catalysis* **2021**, *1*, 1088–1103.
- [47] K. Fan, W. Xie, J. Li, Y. Sun, P. Xu, Y. Tang, Z. Li, M. Shao, *Nat. Commun.* **2022**, *13*, 7958.
- [48] R. Zhang, X. Ma, S. Zhang, H. Cui, C. Li, Y. Wang, Q. Li, C. Peng, Y. Guo, C. Zhi, *Angew. Chem. Int. Ed.* **2025**, *64*, e202507724.
- [49] W. Wang, J. Chen, E. C. M. Tse, *J. Am. Chem. Soc.* **2023**, *145*, 26678–26687.
- [50] W. Liao, J. Wang, G. Ni, K. Liu, C. Liu, S. Chen, Q. Wang, Y. Chen, T. Luo, X. Wang, Y. Wang, W. Li, T.-S. Chan, C. Ma, H. Li, Y. Liang, W. Liu, J. Fu, B. Xi, M. Liu, *Nat. Commun.* **2024**, *15*, 1264.
- [51] Z. Gong, X. Xiang, W. Zhong, C. Jia, P. Chen, N. Zhang, S. Zhao, W. Liu, Y. Chen, Z. Lin, *Angew. Chem. Int. Ed.* **2023**, *62*, e202308775.

Manuscript received: September 21, 2025

Revised manuscript received: November 29, 2025

Manuscript accepted: December 02, 2025

Version of record online: ■■■■■